

Mg/Al layered double hydroxide-Pt nanoparticle composite by delamination-restacking route

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Abstract Platinum nanoparticle intercalated magnesium–aluminum layered double hydroxide (LDH) composite was prepared by restacking the colloidal dispersion of organically modified LDH in the presence of preformed oleyl-amine capped platinum nanoparticles in *n*-butanol. The resultant nanocomposite was characterized using powder X-ray diffraction (PXRD), infrared (IR) spectroscopy, transmission electron microscopy (TEM), selected area electron diffraction and energy dispersive X-ray spectroscopy (EDX). The PXRD pattern of the composite indicates an expanded interlayer due to the intercalation of the Pt-nanoparticles into the LDH interlayer region. PXRD patterns along with the IR spectra confirm the formation of the precursors and the composite. TEM image of Pt-nanoparticles shows spherical particles with an average particle size of ~3 nm and that of the composite shows dispersed Pt-nanoparticles on the LDH matrix. The presence of Pt, Mg and Al in the composite is indicated in the EDX data.

Keywords Layered double hydroxide · Delamination-restacking · Capped Pt-nanoparticles · Nanocomposite

Introduction

Metal nanoparticle supported layered solid composites are gaining immense attention for their application as catalysts (Choudary et al. 2000; Huu et al. 2000, Gérardin et al.

2005), optically tunable materials (Smith et al. 2001), energy conversion devices (Granqvist 2007), sensors (Trudeau 2007; Liu et al. 2012), photocatalysts (Nosaka 2011), etc. Layered solid-nanoparticle composites are generally prepared by impregnation (Komiyama 1985), co-precipitation (Basile et al. 2000), ion-exchange (Rajamathi et al. 2009) and sonochemical (Belova et al. 2008) methods. The composites prepared by the conventional wet impregnation method are not fully reproducible and give rise to an inhomogeneous distribution of the metal nanoparticles on the surface (Komiyama 1985). Composites prepared by co-precipitation in the presence of the noble metals show phase segregation on heating and the size of the metal particles formed were large (Basile et al. 2000). A few composites prepared by growing nanoparticles in the interlayers of the supporting materials show uncontrolled particle growth and an inhomogeneous distribution of the nanoparticles (Xu et al. 2009; Wang et al. 2002). An alternate and more effective way of synthesizing layered solid-nanoparticle composites is by delamination of layered solid to obtain a colloidal suspension of monolayers followed by restacking the layers in the presence of preformed nanoparticles. Nanocomposites prepared starting from monodispersed capped nanoparticles and delaminated layers were found to have uniformly distributed nanoparticles and tunable optical property (Venugopal et al. 2006a). Homogeneity in the distribution of the nanoparticles in the composites occurs as a consequence of the components being mixed at the level of individual delaminated layers.

The method of synthesis, size and distribution of the nanoparticles in a matrix plays an important role in the stability and activity of a catalyst (Peralta et al. 2011; Yeung et al. 1997). The activity is high for smaller sized particles due to large surface area resulting in a larger number of active sites

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(Zhang et al. 2005). The stability of the nanoparticles decreases with their size as they tend to agglomerate to give rise to particles of larger sizes. Capping agents are employed to stabilize the nanoparticles (Amstad et al. 2009) and dispersing nanoparticles on various supports is believed to further stabilize the nanoparticles (Papp et al. 2001). Dispersion of nanoparticles over layered materials is achieved by using delaminated layers as some reports emphasize on two-dimensional layers being used as building blocks to composites through various solution-phase processes (Ma and Sasaki 2010). Delamination or exfoliation of smectites (Walker 1960) and various other layered solids such as chalcogenides (Lerf and Schöllhorn 1977), metal phosphonates (Yamamoto et al. 2001), layered metal oxides (Nazar et al. 1991), layered double hydroxide (LDH) (Pagano et al. 2000), α -metal hydroxides (Nethravathi et al. 2005) and layered hydroxy salts (Rajamathi et al. 2005) have been reported.

Layered double hydroxides are anionic clays with positively charged metal hydroxide layers and charge compensatory interlayer anions. Their general formula is $[M_{1-x}^{2+}M_x^{3+}(\text{OH})_2]^{x+}(A^{n-})_{x/n}$ ($M = \text{Mg}^{2+}, \text{Co}^{2+}, \text{Ni}^{2+}$, etc; $M' = \text{Al}^{3+}, \text{Cr}^{3+}, \text{Fe}^{3+}$, etc; $0.2 \leq x \leq 0.33$; A^{n-} = anions with charge 'n') (Trifiro and Vaccari 1996). LDHs are used as solid-base catalysts (Amstad et al. 2009), catalyst supports (Papp et al. 2001) and precursors to catalysts (Trifiro and Vaccari 1996). LDH-metal composite catalysts are multifunctional materials with metal, acid and basic sites exhibiting metal-support interactions as shown by Kazanski et al. (1997). LDHs with organically modified interlayers were found to undergo delamination in organic solvents like 1-butanol, toluene, formamide, etc. (Pagano et al. 2000; Jobbágy and Regazzoni 2004; Hibino and Jones 2001).

Nanoparticles are known to exhibit various size dependent properties that are different from that of the bulk. As the particle size decreases, the particles are known to show high activity but are quite unstable at the same time. These properties can be exploited only when the nanoparticles are stabilized by some method. Stabilization of surface active nanoparticles by embedding them in layered materials is one of the techniques. Platinum nanoparticles and Pt-based composites are used as catalysts (Anumol et al. 2011), photocatalysts (Taing et al. 2011), electrocatalysts (Habibi and Delnavaz 2010), sensors (Yang et al. 2008), etc. On the other hand, Mg/Al LDH due to its easy availability and tunable composition makes them important supports in various noble-metal based nanocomposites. Among LDHs, Mg/Al-LDHs are interesting due to their wide applications, low toxicity, ease of formation and their economic viability. Generally, LDHs are available in the carbonated form because of ubiquitous nature of CO_2 . Carbonated form of LDH is thermodynamically most stable because of high charge on carbonate ion and the interlayer site symmetry of the LDH matches well with that of the carbonate (D_{3h}

symmetry) ion. But carbonated LDHs do not have many applications due to the inertness of the carbonate anion in the interlayer. In order to make the LDHs application oriented one needs to modify the interlayer with other anions like nitrate, chloride, sulfate or organic anions like acetate, formate, surfactants, etc. Platinum nanoparticles have special properties which sets them apart from the other metal nanoparticles as discussed above. Compositing both, Mg/Al LDH and Pt nanoparticles would synergize the properties of both the materials, thereby making them more application oriented. In this work, we report the synthesis of metal nanoparticle dispersed layered solid composite making use of preformed oleylamine (OA) capped Pt nanoparticles and dodecyl sulfate (DS) modified Mg/Al LDH (hereafter referred to as Pt np-Mg/Al-DS LDH) by delamination–restacking method in *n*-butanol.

Experimental

$\text{Mg}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$, $\text{Al}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$, sodium dodecyl sulfate (SDS; $\text{CH}_3(\text{CH}_2)_{11}\text{OSO}_3\text{Na}$), 25 % ammonia solution (specific gravity = 0.91), toluene, methanol, oleic acid, NaBH_4 and acetone were obtained from Merck India Pvt. Ltd. $\text{H}_2\text{PtCl}_6 \cdot 6\text{H}_2\text{O}$ and oleylamine (OA; $\text{CH}_3(\text{CH}_2)_7\text{CH}=\text{CH}(\text{CH}_2)_8\text{NH}_2$) were procured from Ranchem India Pvt. Ltd. and Sigma Aldrich respectively. All solutions were prepared using Type-II water (specific resistance 15 $\text{M}\Omega \text{ cm}$, Millipore™ Elix-3 water purification system) and dissolved CO_2 was expelled by boiling the water for 10 min.

Preparation of $\text{Mg}_2\text{Al}(\text{OH})_6\text{DS} \cdot 2\text{H}_2\text{O}$ -LDH (Mg/Al-DS LDH)

Dodecyl sulfate (DS) intercalated LDH, $\text{Mg}_2\text{Al}(\text{OH})_6\text{DS} \cdot 2\text{H}_2\text{O}$ (hereafter referred to as Mg/Al-DS LDH) was prepared by coprecipitation. In a typical experiment, metal nitrate mixture of Mg^{2+} and Al^{3+} with a molar ratio of 2:1 was added drop wise into 50 ml 1 M ammonia solution which contains six times excess of the stoichiometric amount of SDS with continuous stirring. The slurry was aged for 18 h at 65 °C in an air oven. The solid was washed free of ions with hot decarbonated water six to seven times, twice with acetone and dried at 65 °C overnight.

Preparation of oleylamine capped platinum nanoparticles (OA-capped Pt-nps)

OA-capped Pt-nps were prepared by a method reported elsewhere with slight modifications (Halder and Ravishanker 2006). A mixture of 500 mg of $\text{H}_2\text{PtCl}_6 \cdot 6\text{H}_2\text{O}$,

660 ml toluene, 5 ml oleic acid was refluxed for 6 h and left undisturbed for 16 h. 192 mg of NaBH_4 in 40 ml methanol was added drop wise to the reaction mixture. The solution was allowed to stand at room temperature for 1 hour. Additional 20 mg of NaBH_4 was added directly to the reaction mixture and was left undisturbed overnight. The brownish-black solid formed was separated by centrifugation, washed with acetone (3–4 times) and dried in air at 65 °C.

Preparation of oleylamine Pt nanoparticles-Mg/Al-DS LDH (Pt-np-Mg/Al-DS LDH) composite

100 mg of Mg/Al-DS LDH was delaminated in 150 ml 1-butanol by sonicating (35 kHz) at 70 °C for 2 h. Simultaneously, 10 mg of OA-capped Pt-nps were dispersed in 50 ml of 1-butanol by sonication at 70 °C for 30 min. The two dispersions were mixed and sonicated for another hour. The solvent was evaporated using a rotatory evaporator at 80 °C. The solid composite thus obtained was washed with acetone to remove remnant butanol and was dried in an air oven at 65 °C.

Characterization

The powder X-ray diffraction (PXRD) patterns of the samples were recorded on a Philips X'Pert Pro diffractometer fitted with secondary graphite monochromator ($\text{CuK}\alpha$ radiation, $\lambda = 1.5418 \text{ \AA}$). The data was collected at the rate of 2° 2θ per minute over the 2θ range $2\text{--}70^\circ$. Infrared (IR) spectra of the samples were collected using Nicolet IR200 FT-IR spectrometer (KBr pellet method, 4 cm^{-1} resolution). The composite and the platinum nanoparticles were also characterized using Tecnai F30 transmission electron microscopy (TEM) interfaced with energy dispersed X-ray spectroscopy (EDS). Samples for TEM were prepared by dispersing a small amount of the solid in about 10 ml of acetone by sonication, a drop of which was spotted on a carbon coated grid and dried at room temperature.

Results and discussions

Figure 1a shows the PXRD pattern of the as synthesized oleylamine capped Pt-nanoparticles in which the (111), (200) and (220) reflections of platinum were seen at 2θ values of 40.08° ($d = 2.265 \text{ \AA}$), 46.45° ($d = 1.962 \text{ \AA}$) and 67.75° ($d = 1.1826 \text{ \AA}$), respectively (ICSD Number 64917). The structure of platinum nanoparticles corresponds to face-centered cubic with a space group of $Fm\bar{3}m$ (Wyckoff 1963). Furthermore, the average particle size of the oleylamine capped Pt-nanoparticles calculated using

Scherrer's equation was found to be $\sim 5.5 \text{ nm}$. Figure 1b shows the PXRD pattern of dodecyl sulfate anion (DS^-) intercalated Mg/Al-LDH. The interlayer distance of Mg/Al-DS LDH (calculated from the position of the first basal reflection) was found to be 23.9 \AA indicating the presence of DS^- anion in the interlayer region of the LDH with the alkyl chains of the surfactant arranged parallel to the crystallographic c axis (Guo et al. 2005) and perpendicular to the layers. In addition to the basal reflections, sawtooth-shaped reflections of the ($h0l$) and ($0kl$) planes were also seen at 2θ value in between $35\text{--}36^\circ$ and $\sim 61^\circ$ indicating turbostraticity in the layered solids (Warren and Bodenstein 1965). The PXRD pattern of Pt-np-Mg/Al-DS LDH composite (Fig. 1c) shows reflections due to both the LDH and Pt-nanoparticles (reflections due to platinum nanoparticles are visible in the enlarged inset). The interlayer distance of the Pt-np-Mg/Al-DS LDH composite was found to be 27.6 \AA . The increase in the interlayer distance of the composite when compared to the Mg/Al-DS LDH precursor is due to the incorporation of the Pt-nanoparticles into the interlayer region of the LDH, which is in line with an earlier report that showed the increase in the interlayer distance due to incorporation of CdSe nanoparticles (Venugopal et al. 2006a). The increase in the interlayer distance of the composite due to the intercalation of the OA-capping agent as the positively charged OA-surfactant ions (excluding the Pt-nanoparticle entity) are ruled out as it is unfavorable. The increase in the interlayer distance of

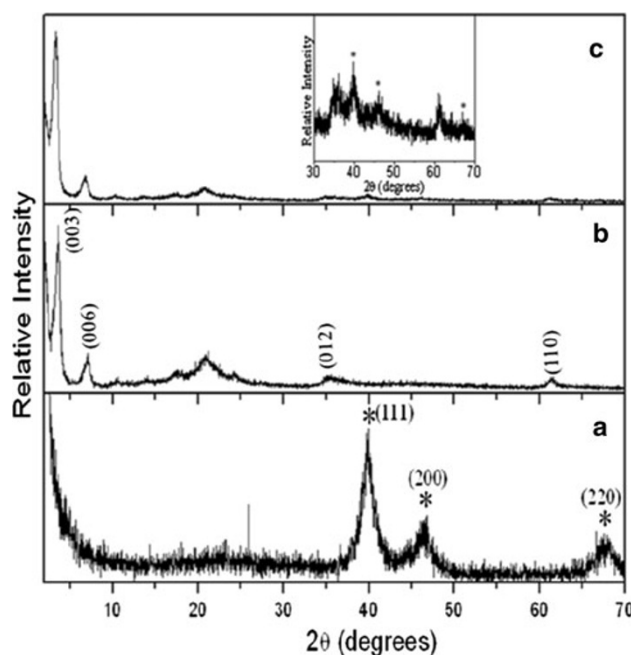


Fig. 1 PXRD pattern of OA-capped Pt-nps (a), Mg/Al-DS LDH (b) and Pt-np-Mg/Al-DS LDH composite (c). Inset shows an enlarged portion of c from 2θ 30° to 70° . Reflections due to Pt-nanoparticles are marked with asterisk

the composite due to swelling also can be ruled out as the samples were dried in an air oven at 65 °C to constant weight driving out the intercalated solvent molecules. Restacking of surfactant intercalated LDHs has been standardized in various solvents and the *d*-spacing of the LDHs got after restacking was found to be very close to the parent LDHs as reported by Venugopal et al. (2006b).

Figure 2 shows the IR spectra of Mg/Al-DS LDH, OA-capped Pt-nps and Pt-np-Mg/Al-DS LDH composite. The IR spectrum of Mg/Al-DS LDH (Fig. 2a) shows bands due to the stretching and bending modes of hydrogen bonded –OH groups of the hydroxide slabs and the adsorbed and intercalated water molecules at $\sim 3,500$ and $\sim 1,640$ cm^{-1} respectively. Bands due to C–H stretching vibrations of the surfactant alkyl chains are seen at 2,853, 2,919 and 2,950 cm^{-1} suggesting an all-trans conformation (Snyder et al. 1982; MacPhail et al. 1984) and a band due to S=O stretching vibrations at 1,222 cm^{-1} of sulfate polar head of the surfactant were also observed. The all-trans conformation of the surfactant alkyl chain suggests that these chains are linear without bends and twists and this correlates well with the basal spacing observed in the XRD pattern (Fig. 1b). The IR spectrum of the OA-capped Pt-nps (Fig. 2b) showed bands due to NH stretching and bending modes at 3,435 and 1,600 cm^{-1} , respectively. The bands due to C–H stretching vibrations of the oleylamine chain appeared between 2,829 and 2,980 cm^{-1} . The IR spectrum of the composite (Fig. 2c) is identical to that of Mg/Al-DS LDH except for a slight broadening at around

3,500 cm^{-1} caused due to the overlapping of the O–H and N–H stretching bands.

Figure 3a, b shows bright field TEM images of OA-capped Pt-nps and Pt-np-Mg/Al-DS LDH composite, respectively. The TEM image of Pt-nanoparticles showed spherical particles with an average particle size of 3 nm, which is much lower than the values calculated using Scherrer's expression. This discrepancy in particle size arises as particle sizes measured by XRD (using Scherrer's equation) can be easily overestimated by even a small fraction of larger particles, which will contribute to a larger extent to the observed signal since the average measured by XRD is in fact weighted by the particle volume. TEM of the composite showed uniform distribution of Pt nanoparticles over the LDH matrix and the layered structure of the Mg/Al-LDH is intact. Selected area electron diffraction (SAED) patterns of the platinum nanoparticles (Fig. 3c) show characteristic features of platinum (Ha et al. 2011) and the indices are as mentioned in the figure. The SAED pattern of the composite (Fig. 3d) showed diffuse rings due to platinum along with the bright spots due to Mg/Al LDH that appears as a distorted hexagon (Abdelouas et al. 1994). The presence of Pt-nanoparticles in the Mg/Al-LDH matrix is further supported from the energy dispersive X-ray spectrum (EDS) of the Pt-np-Mg/Al-DS LDH composite shown in Fig. 4. Peaks due to platinum, magnesium and aluminum are seen and peaks due to copper are due to the grid on which the sample was coated.

When the LDH is dispersed in 1-butanol, the solvent molecules enter the interlayer region leading to high degree swelling and delamination of the metal hydroxide layers. The delaminated colloidal dispersion is mixed with pre-formed oleylamine capped Pt-nanoparticles and the LDH layers are restacked along with Pt-nanoparticles by solvent evaporation to get a composite that is evident from the PXRD pattern (Fig. 1c). An enlarged inset clearly legitimates reflections due to Pt nanoparticles present along with the 'ab' plane reflections of the LDHs. Use of delaminated layers for composite preparation is essential as it facilitates easy mixing of the components yielding composite with uniform distribution of nanoparticles in the LDH matrix as evident in the TEM image of the composite (Fig. 3b). The incorporation of the Pt nanoparticles into the interlayer regions of the LDH is supported by the increase in the average interlayer distance of the composite by ~ 4 Å. A very small increase in the average interlayer distance of the composite when compared to that of the pristine LDH is due to very small amounts of Pt nanoparticles (1:10 weight ratio) used for composite preparation. Increase in the amount of the nanoparticles in such a composite leads to a corresponding increase in the interlayer distance of the layered solid has already been reported. Moreover, the differently shaded Pt nanoparticles in the TEM of the composite (Fig. 3b) is due to

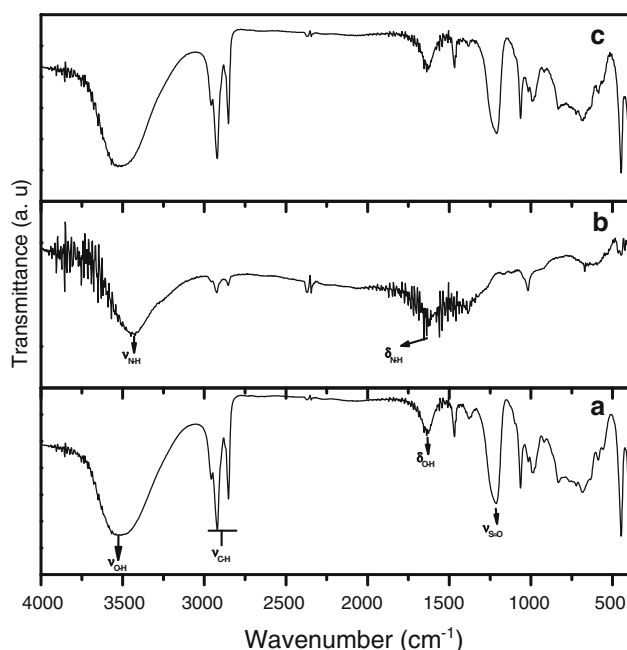


Fig. 2 IR spectra of Mg/Al-DS LDH (a), OA-capped Pt-nps (b) and Pt np-Mg/Al-DS LDH composite (c)

Fig. 3 Bright field TEM images and SAED pattern of OA-capped Pt-nps (**a, c**) and Pt-np-Mg/Al-DS LDH composite (**b, d**)

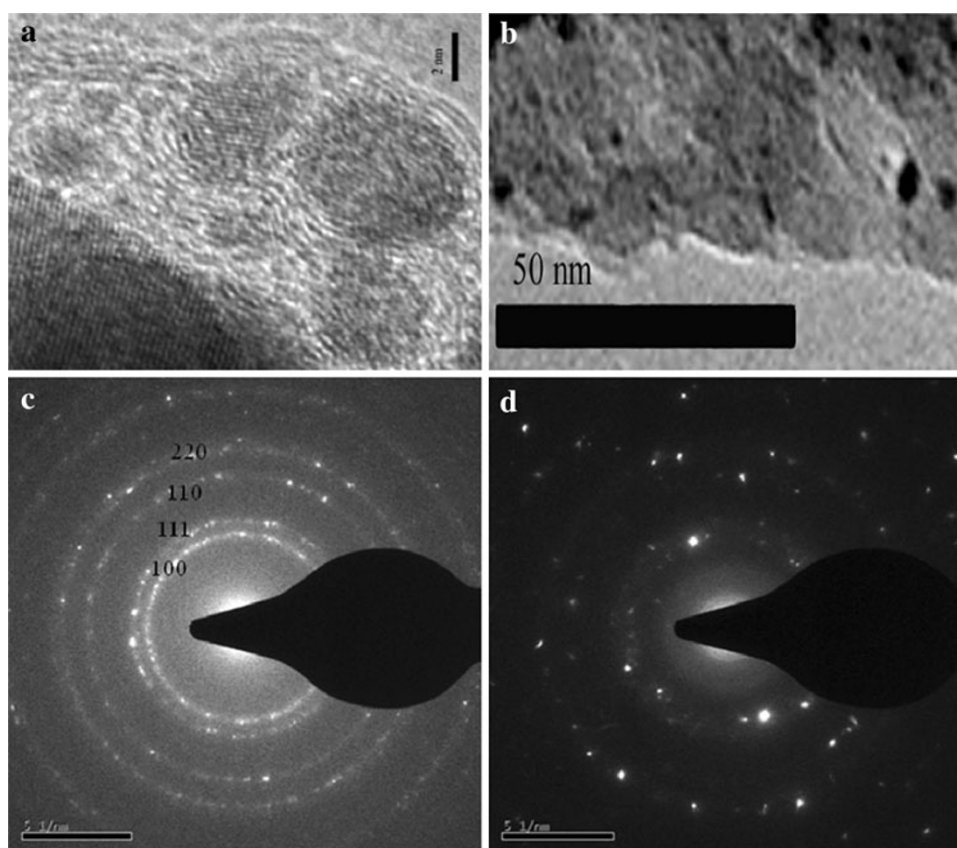
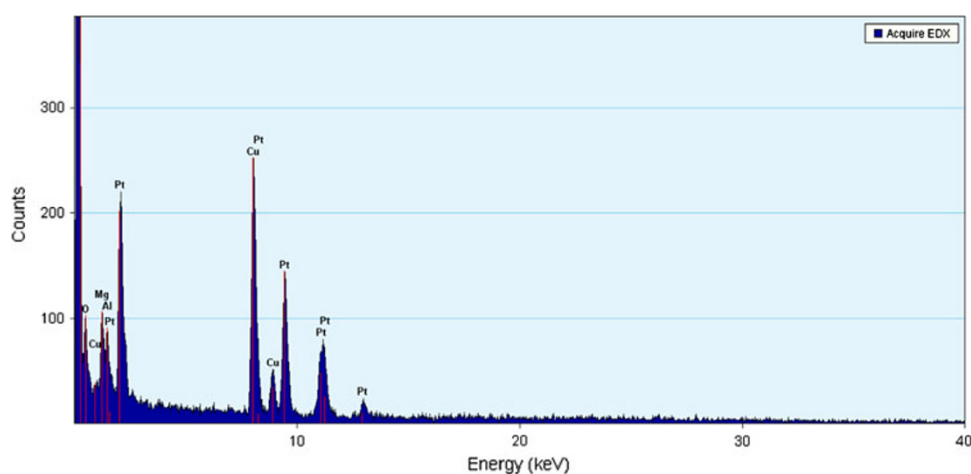


Fig. 4 Energy dispersive X-ray spectrum of Pt-np-Mg/Al-DS LDH composite



the presence of the Pt nanoparticles present at various depths in the un-exfoliated LDH matrix. The technique employed for the synthesis of the composite can be used for other than Mg/Al LDH systems and also for other nanoparticles to suit particular application/s.

Conclusion

A general route for synthesis of noble metal nanoparticle intercalated layered solid composite was proposed taking

the synthesis of Mg/Al-DS LDH supported Pt-nanoparticle composite as a specific example. The composite was prepared by restacking the delaminated Mg/Al-DS LDH in the presence of preformed oleylamine capped Pt-nanoparticles in 1-butanol. The delamination-restacking technique employed can be applied to other layered solid-nanoparticle systems of various compositions.

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